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Electrocatalytic performance and cell voltage characteristics of 1st-row transition metal phosphate (TM-Pi) catalysts at neutral pH

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Abstract

Water electrolysis represents a clean and sustainable route for large-scale hydrogen generation. However, efficient water splitting is hindered by the kinetically sluggish oxygen evolution reaction (OER), which requires significant energy inputs to drive the reaction at sufficiently fast rates. Recently, an increasing number of applications have emerged that require water electrolysis at neutral pH and under ambient conditions. This requirement creates additional challenges as the electrolysis of water is favorable in acidic and alkaline conditions. In order to tackle these challenges, considerable efforts have been devoted to the development of earth-abundant, highly effective, and robust electrocatalysts for the OER at pH=7. Of these catalysts, amorphous transition-metal phosphates have attracted wide attention because of their unique electrocatalytic properties. In this paper, the OER performance of a series of amorphous first-row transition metal phosphate (TM-Pi) catalysts, namely Co-Pi, NiFe-Pi and Fe-Pi prepared with different deposition strategies onto various substrates, is comparatively studied in a neutral phosphate buffer solution (PBS). Additionally, a simplified cell model is applied to analyze the current-voltage characteristics and quantitatively evaluate and compare the reversible, ohmic, and activation overvoltage components of the studied TM-Pi. It is found that TM-Pi catalysts deposited onto a highly ordered nickel foam (NF) substrate are competitive with commercial Pt and IrO₂ catalysts in terms of OER activity and long-term stability.

Keywords: In situ water electrolysis, electrocatalyst, transition metal phosphate, oxygen evolution reaction, pH-neutral electrolyte

1. Introduction

The intensive use of fossil and nuclear energy has exacted a considerable toll on the natural environment and has led to rapid depletion of many natural resources, adverse climate impacts from greenhouse gas emissions, as well as water, air and land pollution. Growing public awareness of climate change and other environmental issues have resulted in demands for clean and sustainable technologies based on renewable energy [1, 2, 3]. Hydrogen, which is the simplest and lightest element of the periodic table, has a high energy density and is considered an eco-friendly fuel to meet the world's increasing energy demand. Currently, the major part of hydrogen production is based on fossil fuel refining processes, such as steam reforming or partial oxidation of methane, which release significant amounts of carbon dioxide into the atmosphere. In contrast, electrochemical water splitting based on renewable electricity provides a sustainable carbon-neutral route for storing energy from intermittent sources using hydrogen as a carrier [4, 5].

The water electrolysis process has formed the basis for a number of different energy storage or conversion devices, of which the most efficient and robust are electrolyzers [6, 7], fuel cells [8], and metal-air batteries [9]. Bio-electrochemical systems (BES) [10] and hybrid biological-inorganic (HBI) systems [11] represent a fast-developing area of research and are interesting examples of integrating water electrolysis into the biological process to fix carbon dioxide into multicarbon organic compounds. Even though the module design of such systems varies, a common factor is that they are all two-electrode systems involving the hydrogen evolution reaction (HER) or oxygen reduction reaction (ORR) at the cathode and water oxidation or the oxygen evolution reaction (OER) at the anode [12]. The common constraint of these systems is the sluggish kinetics of the oxygen evolution reaction (OER) in which molecular oxygen is formed through a complex pathway involving the extraction of four electrons and four protons. The hydrogen evolution reaction (HER) however is exclusively a two-electron transfer reaction and, therefore, requires substantially less energy (overpotential) to overcome the kinetics of the reaction [12, 13].

The oxygen evolution reaction (OER) is highly pH-dependent. In alkaline conditions, four hydroxyl groups (OH^-) are oxidized into two water molecules (H_2O) and one molecule of oxygen (O_2), while in acidic and neutral conditions, two water molecules (H_2O) are oxidized to be transformed into four protons (H^+) and one oxygen molecule (O_2) [13]. Electrochemical water split-

38 ting is hindered by the high stability of water, which requires 237 kJ mol^{-1}
39 of Gibbs free energy change for the overall reaction [14]. According to the
40 thermodynamics, the equilibrium or reversible cell voltage, which is the low-
41 est potential required for the electrolysis to take place at 25°C and 1 atm,
42 is equal to 1.23 V. In practice, however, commercial electrolyzers typically
43 operate in the range of 1.8–2.1 V, indicating that water electrolysis proceeds
44 far from its equilibrium potential [15]. Significant energy losses caused by the
45 large overpotentials required to drive water splitting at sufficiently fast rates
46 are mainly the result of the kinetically sluggish oxygen evolution reaction
47 (OER) [12, 13, 16]. Therefore, highly efficient OER catalysts are required
48 to lower the energy barrier and increase the overall efficiency of the water
49 splitting process.

50 At the present time, noble metal-based catalysts made of ruthenium (Ru)
51 and iridium (Ir) and especially their oxides exhibit superior performance for
52 OER and are considered benchmark catalysts. A major drawback of these
53 materials is their unsuitability for large-scale applications owing to their high
54 cost and scarcity. In addition, in both acidic and alkaline electrolytes, IrO_2
55 and RuO_2 experience stability issues at high anodic potentials because of
56 their oxidation into IrO_3 and RuO_4 , respectively, and further dissolution in
57 the electrolyte [12]. Considerable efforts have been therefore made to de-
58 velop active, stable and low-cost catalysts made of earth-abundant materials
59 capable of operating at low overpotentials at different pHs as substitutes for
60 precious metal catalysts [12, 13].

61 Various transition metal (TM) oxides [17, 18, 19], hydroxides [20], sul-
62 fides [21, 22], nitrides [23, 24], borides [25, 26], carbides [27, 28], selenides
63 [29] and phosphides [30, 23, 31, 32, 33] have been extensively investigated
64 as efficient catalysts for overall water splitting. Transition metal phosphates
65 (TM-Pi) have attracted widespread scientific interest after the development
66 of a cobalt phosphate (Co-Pi) catalyst by Matthew W. Kanan and Daniel
67 G. Nocera in 2008 [34]. In subsequent studies, numerous other TM-Pi cata-
68 lysts have been developed, and their electrocatalytic activity towards OER
69 and HER has been shown. A common feature of all these catalysts is the
70 doping of phosphate groups into a crystal lattice of transition metals, such
71 as Ir [16, 35], Fe [36, 37], Co [34, 35], Ni [38, 39, 40, 41] and Mn [42] by dif-
72 ferent methods including electrodeposition, precipitation and hydrothermal
73 synthesis. Phosphate groups facilitate the oxidation of metal atoms during
74 the proton-coupled electron transfer process and distort the native atomic
75 geometry favoring adsorption and oxidation of water molecules [43]. As a

76 result of this unique feature, several reported catalysts, such as Ir–Pi and
77 IrCo–Pi, exhibit an intrinsic performance superior to the benchmark IrO₂
78 [16, 35]. However, even though the content of Ir in IrCo–Pi has been reduced
79 by 50%, there is still a considerable capacity for optimization and reduction
80 in expensive Ir usage.

81 In the present paper, a series of first-row transition metal phosphates
82 (TM-Pi) are prepared and tested in a neutral phosphate buffer solution
83 (PBS). It is well known that water electrolysis is kinetically difficult in pH-
84 neutral conditions and more favorable in basic and acidic conditions because
85 of the deprotonated molecules available for oxygen and hydrogen evolution
86 reactions [14]. However, pH-neutral conditions exert a significantly less corro-
87 sive effect on the electrochemical devices and are vital for emerging biological
88 processes with integrated water electrolysis. The materials analyzed included
89 Co–Pi, Fe–Pi and NiFe–Pi. Co–Pi was selected as a benchmark catalyst
90 from the TM-Pi family, which was exclusively developed to perform water
91 splitting in artificial photosynthesis systems at neutral pH under ambient
92 conditions [34]. Iron (Fe) is the cheapest and most abundant TM in the
93 Earth’s crust. Therefore, Fe–Pi prepared by a simple and cheap successive
94 ionic layer deposition and reaction (SILAR) method [36, 37] was selected as
95 a candidate for analysis. Literature review suggests that Ni–Fe catalysts
96 provide a promising alternative to catalysts based on precious metals. It is
97 noteworthy that Ni–Fe catalysts are considerably more active for OER than
98 pure Ni or Fe [39, 41]. Consequently, bimetallic NiFe–Pi [39] was selected
99 to quantify the synergistic effect of iron in NiFe-based structures. The per-
100 formance of Fe–Pi and NiFe–Pi has been extensively studied in an alkaline
101 medium, but to the authors’ knowledge, it has not been studied in detail
102 in neutral conditions. **The main contribution of the present study is the
103 comparative analysis of TM-Pi catalyst on different substrates under neu-
104 tral conditions. A simplified mathematical electrolyzer cell model is used
105 to deconvolute different overpotentials based on measured current–voltage
106 characteristics.**

107 **2. Experimental section**

108 *2.1. Materials*

109 Analytical grade reagents NiSO₄ · 6 H₂O, FeSO₄ · 7 H₂O, NaH₂PO₂, NaOAc · 3 H₂O,
110 KH₂PO₄, K₂HPO₄, Co(NO₃)₂ · 6 H₂O, FeCl₃ · 6 H₂O, ethanol, and acetone

111 were purchased from Sigma-Aldrich, USA. All chemicals were used as re-
112 ceived without additional purification. Graphite electrodes were acquired
113 from Gamry Instruments, USA. Titanium (Ti) coated with IrO_2 , and Pt
114 catalysts were purchased from PV3 Technologies, UK. Nickel foam (NF,
115 thickness: 2mm, porosity: 97%, purity: 99.8%, PPI: 110) was acquired from
116 Tmax Battery Equipments Limited Company, China.

117 2.2. Catalyst fabrication (*NiFe-Pi*, *Co-Pi*, and *Fe-Pi*)

118 The procedure for catalyst fabrication was adopted from previous reports
119 [34, 37, 39]. Electrodeposition and successive ionic layer deposition and reac-
120 tion (SILAR) were the methods used for catalyst fabrication. The electrode-
121 position was carried out in the experimental setup depicted in Fig. 1. The
122 setup consisted of: (i) a three-electrode electrolyzer cell with a cross-sectional
123 area of 2.6 cm^2 ; (ii) a WaveNow potentiostat to conduct electrochemical mea-
124 surements; (iii) a constant flow pump to circulate the electrolyte through the
125 external vessel equipped with the temperature measurement; and (iv) a water
126 bath with an immersed Lauda heater to maintain constant temperature.

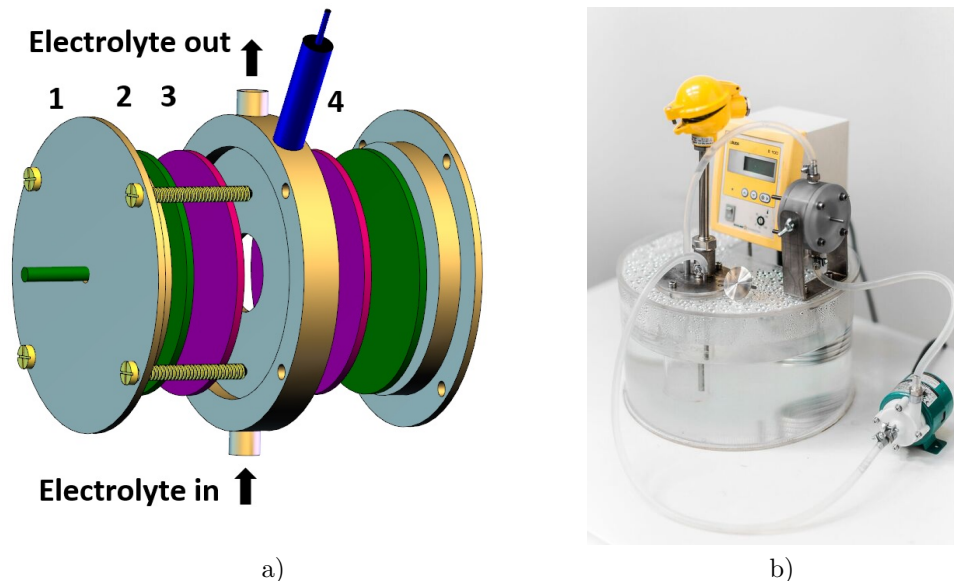


Fig. 1. Experimental setup used for the electrolysis tests. (a) 3D model of the electrolyzer cell; 1: teflon frame, 2: conductive stainless steel plates, 3: working and counter electrodes, 4: reference electrode (Ag/AgCl). (b) Photo of the experimental setup.

127 Graphite and nickel foam (NF) were used as substrates for deposition
128 of the catalysts. Prior to deposition, a circular piece of NF (15.2 cm^2) was
129 cleaned in 2M HCL to remove the nickel oxide layer. Subsequently, ultrasonic
130 cleaning was performed in ethanol, acetone and distilled deionized water to
131 complete removal of surface contaminants. Pretreatment of the graphite
132 substrates included polishing with sandpaper and rinsing with acetone and
133 distilled deionized water. All samples were dried in ambient air prior to the
134 deposition of the TM-Pi catalysts.

135 The NiFe-Pi catalyst was prepared with a cathodic electrodeposition
136 strategy by cyclic voltammetry (CV) in the potential range from -1.2 to -
137 0.7 V (vs.Ag/AgCl) at a scan rate of 5 mV/s for 8 cycles. The aqueous
138 source solution contained 0.0125, 0.025, 0.0375 M $\text{NiSO}_4 \cdot 6\text{H}_2\text{O}$, 0.0125,
139 0.025, 0.0375 M $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$, 0.5 M NaH_2PO_2 , and 0.1 M $\text{NaOAc} \cdot 3\text{H}_2\text{O}$
140 dissolved in distilled deionized water. The Co-Pi catalyst was prepared
141 with an anodic electrodeposition strategy by CV at a scan rate of 5 mV/s
142 for 100 cycles. Potential ranges from 0.9 to 1.2 V (vs.Ag/AgCl) and from
143 0.8 to 1.1 V (vs.Ag/AgCl) were used when depositing onto graphite and NF,
144 respectively. To obtain the solution for the electrodeposition, 0.5 mM of
145 $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ was added to the phosphate buffer solution (PBS), which
146 was prepared by mixing 0.1 M KH_2PO_4 and 0.1 M K_2HPO_4 . In both elec-
147 trodeposition methods, graphite and NF were used as the working electrodes
148 (WE), while a stainless steel plate was used as the counter electrode (CE).
149 For the SILAR deposition method, 10 mM aqueous solutions of KH_2PO_4
150 and $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ were prepared separately as the sources of cations and an-
151 ions, respectively. To prepare the Fe-Pi coating, the cleaned substrates were
152 successively immersed first in a KH_2PO_4 solution for 10 s and then imme-
153 diately in a $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ solution for 10 s. This procedure was repeated
154 for 10 min. All solutions were prepared in ultrapure doubly distilled water
155 obtained from a PURELAB flex system. After deposition, all samples were
156 again gently rinsed with distilled deionized water and dried in ambient air
157 before the electrochemical measurement tests.

158 *2.3. Material characterization*

159 A Hitachi S-3400N field-emission scanning electron microscope (SEM)
160 equipped with energy dispersive X-ray spectroscopy (EDX) measurement was
161 used to examine the surface morphologies and the compositional distribution
162 of the elements of the fabricated catalysts. SEM images were obtained by

163 operating the microscope at 10 kV and 20 mA using a UDV (secondary elec-
 164 tron detector) and a BSE (backscatter electron detector). X-ray diffraction
 165 (XRD) analysis performed with a Bruker D8 Advance X-ray diffractometer
 166 was used to analyze the crystal structure of the studied TM-Pi catalysts.
 167 XRD patterns were obtained at 40 kV, 30 mA with Cu K α -type radiation.

168 2.4. Electrochemical measurements

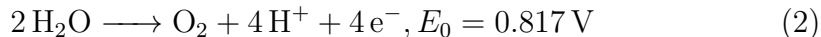
169 All the electrochemical measurements were conducted with a WaveNow
 170 Potentiostat from Pine Research Instrumentation, USA, in the three-electrode
 171 system shown in Fig. 1. As-prepared TM-Pi catalysts deposited on NF or
 172 graphite were directly used as the working electrodes (WE) without further
 173 treatment. A titanium (Ti) coated with a Pt circular electrode plate and an
 174 Ag/AgCl, Cl $^-$ (3M) electrode were used as the counter electrode (CE) and the
 175 reference electrode (RE), respectively. Polarization curves were obtained by
 176 linear sweep voltammetry (LSV) at a scan rate of 5 mV s $^{-1}$ in a PBS solution,
 177 which was prepared by mixing 0.1 M KH $_2$ PO $_4$ and 0.1 M K $_2$ HPO $_4$. Long-
 178 term durability tests were carried out in the chronopotentiometry mode for
 179 36 h in a PBS. All the potentials measured in this work were calibrated to a
 180 reversible hydrogen electrode (RHE) using the following equation:

$$E_{\text{RHE}} = E_{\text{Ag/AgCl}} + 0.1976 \text{ V} + 0.059 \text{ pH} \quad (1)$$

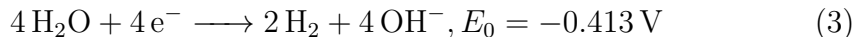
181 2.5. Cell model

182 Water electrolysis in neutral conditions (pH=7) is described by the fol-
 183 lowing electrochemical reactions [14].

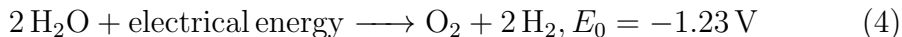
The oxidation half-reaction at the anode—oxygen evolution reaction (OER):



The reduction half-reaction at the cathode—hydrogen evolution reaction (HER):



The overall reaction in the electrolytic cell:



184 Virtually, higher voltages than equilibrium or reversible voltage (1.23 V)
 185 are required to initiate the dissociation of water. The high overvoltage and

186 the subsequent energy loss are mainly caused by the additional overvoltages
 187 presented in the following equation [15]:

$$U_{\text{cell}} = U_{\text{rev}} + U_{\text{ohm}} + U_{\text{act}} + U_{\text{con}}, \quad (5)$$

188 where U_{cell} is the cell voltage, U_{rev} is the reversible open circuit voltage,
 189 U_{ohm} is the overvoltage caused by ohmic losses in the cell elements, U_{act}
 190 is the activation overvoltage caused by electrode kinetics, and U_{con} is the
 191 concentration overvoltage caused by mass transport processes (usually, U_{con}
 192 is much lower than U_{ohm} and U_{act} , and it can thus be neglected).

193 A water electrolysis model developed and validated in our previous studies
 194 [44, 45] was applied to analyze the collected current–voltage characteristics.
 195 The simplified cell model presented in the equation below allows quantitative
 196 evaluation and comparison of reversible, ohmic (the 2nd term in the equation
 197 after the equal sign) and activation overvoltages (the 3rd term in the equation
 198 after the equal sign) of different electrode sets.

$$U_{\text{cell}} = U_{\text{rev}} + \frac{\delta_{\text{m}} i_{\text{cell}}}{\sigma_{\text{m}}} + \alpha \operatorname{arcsinh} \left(\frac{i_{\text{cell}}}{2i_0} \right), \quad (6)$$

199 where δ_{m} is the distance between the electrodes (expressed in cm), i_{cell} is the
 200 current density (A cm^{-2}), σ_{m} is the conductivity of the electrolyte (S cm^{-1}),
 201 α is the charge transfer coefficient, and i_0 is the exchange current density on
 202 the electrode surfaces. U_{rev} , σ_{m} , α , and i_0 are the parameters to be fitted by
 203 the experimental data.

204 3. Results and discussion

205 3.1. TM-Pi catalyst formation

206 Typical CV curves of the NiFe–Pi and Co–Pi electrodeposition process
 207 are presented in Fig. 2. The CV approach ensures fast deposition of TM
 208 ions in different potential ranges, and it is beneficial in comparison with
 209 traditional potentiostatic deposition of TM-Pi catalysts. It can be clearly
 210 seen from the curves that there is a steep increase in the voltammetric current
 211 observed over the first 20–30 cycles and a subsequent gradual increase during
 212 the electrodeposition of Co–Pi. A similar behavior can be observed in the
 213 potential region of the HER; the voltammetric current decreases fast at the
 214 beginning of the deposition of NiFe–Pi, and slows down at the end. It would
 215 appear that the increasing coverage of the substrate surface and the rising

216 thickness of the TM-Pi coating accelerate the rate of the electrolysis reaction
 217 in both cases. After deposition of NiFe-Pi and Co-Pi, the central part of
 218 the NF substrate was evenly coated with a dark coating. When depositing
 219 NiFe-Pi onto graphite, the surface coating had a silver color as seen in Fig. 3.

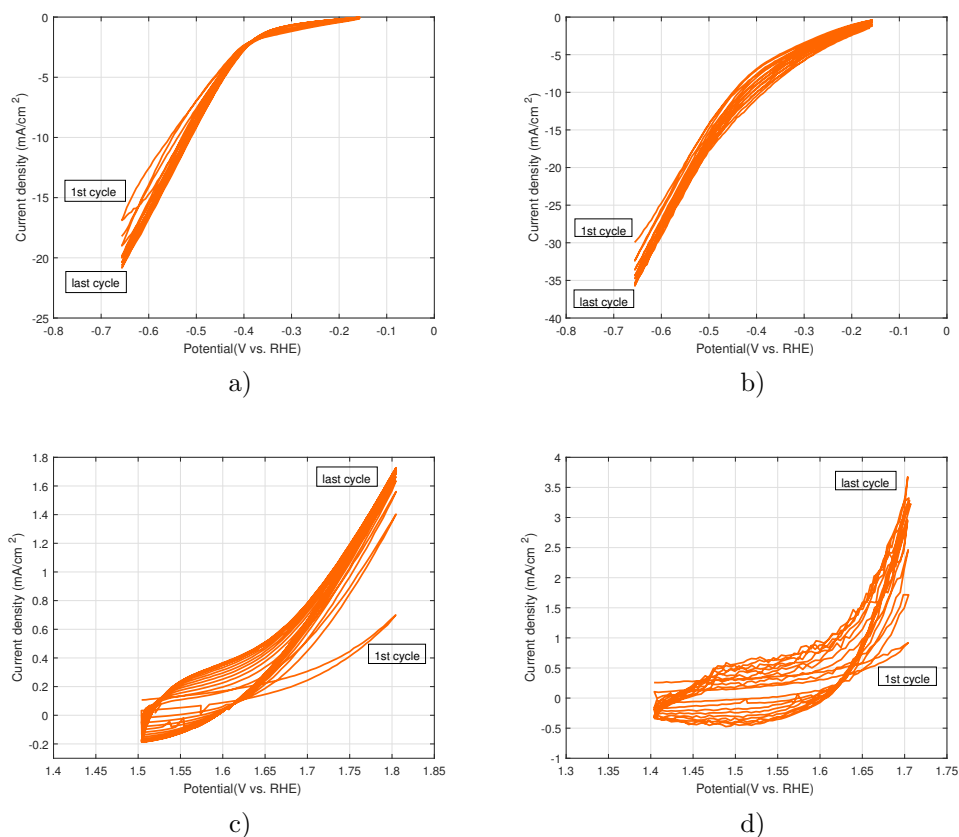


Fig. 2. Cyclic voltammograms of the deposition process of (a) NiFe-Pi onto graphite, (b) NiFe-Pi onto NF, (c) Co-Pi onto graphite and (d) Co-Pi onto NF.

220 The SILAR deposition method to prepare Fe-Pi is considered an inexpensive
 221 and facile strategy. Successive immersion of the substrate in the 10 mM
 222 cation solution of KH_2PO_4 and the 10 mM anion solution of $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$
 223 for 10 min resulted in the formation of a white flocculent precipitate all over
 224 the surface which could be easily detected with the naked eye.

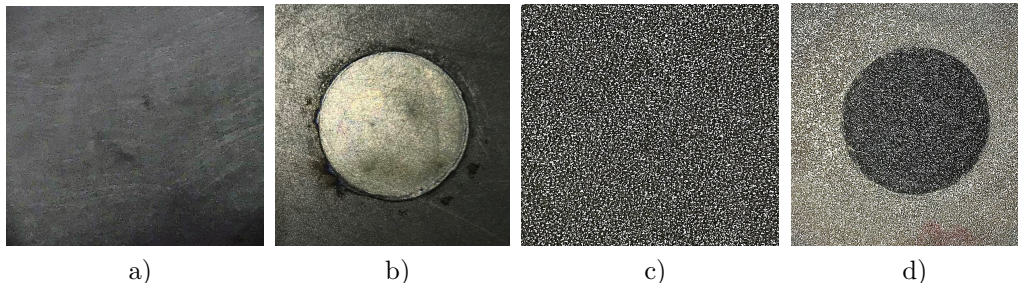


Fig. 3. Photos of electrode surfaces: (a) bare graphite, (b) graphite coated with NiFe-Pi, (c) NF, and (d) NF coated with Co-Pi.

225 *3.2. Physical characterization*

226 XRD patterns were recorded for the substrate prior to and after the catal-
 227 yst deposition. A typical XRD diagram of the TM-Pi catalysts deposited
 228 on NF is presented in Fig. 4. Diffraction peaks were observed at 44.5° , 55.8° ,
 229 and 76.4° for the NF substrate. The peaks indicates that all the synthesized
 230 TM-Pi catalysts are amorphous in nature.

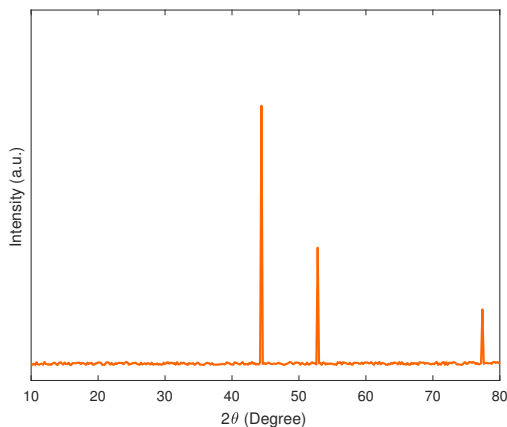


Fig. 4. Typical XRD diagram of the TM-Pi catalysts deposited on NF.

231 The morphology of the TM-Pi catalysts was studied using SEM. Fig. 5
 232 shows SEM images of bare NF and as-deposited Fe-Pi and Co-Pi, while Fig. 6
 233 depicts the surface morphology and elemental mapping of NiFe-Pi deposited
 234 on the NF substrate. The SEM analysis revealed that the surface of the NF
 235 was completely and uniformly coated with TM-Pi catalysts in all cases. The

236 example of elemental mapping in Fig. 6d shows uniform distribution of Ni, Fe,
 237 and P in the film of NiFe-Pi catalysts. This observation applied also to other
 238 TM-Pi catalysts. According to SEM images at high magnification, all TM-Pi
 239 catalysts have different surface structures. In particular, the Fe-Pi coating
 240 demonstrates a highly rippled flake structure with numerous agglomerates.
 241 A Co-Pi layer is formed on the surface of the NF substrate with numerous
 242 3D-porous micrometer-size particles on top of the film. Hierarchical NiFe-
 243 Pi mainly consists of nanobubbles with sizes ranging from approximately
 244 200 nm to 500 nm. Cracks observed in all the TM-Pi catalysts at higher
 245 magnifications probably result from loss of water when drying of samples in
 246 ambient air.

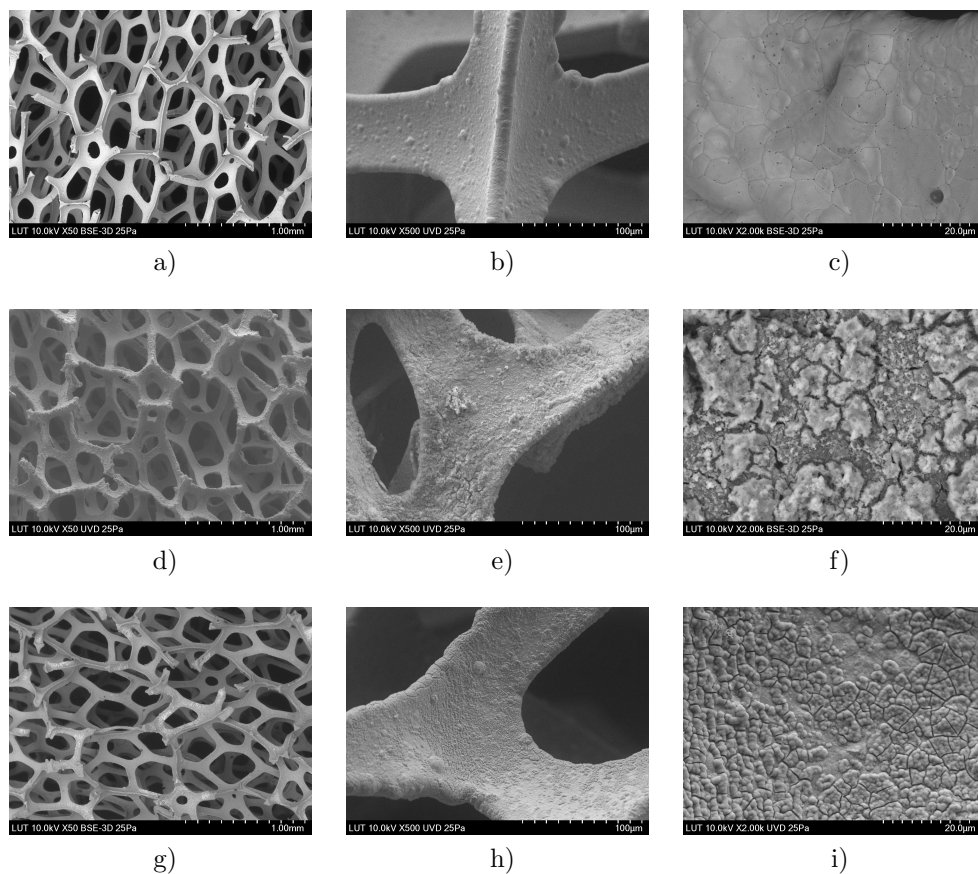


Fig. 5. SEM images of (a–c) bare nickel foam (NF), (d–f) Fe-Pi on NF and (g–i) Co-Pi on NF at different magnifications.

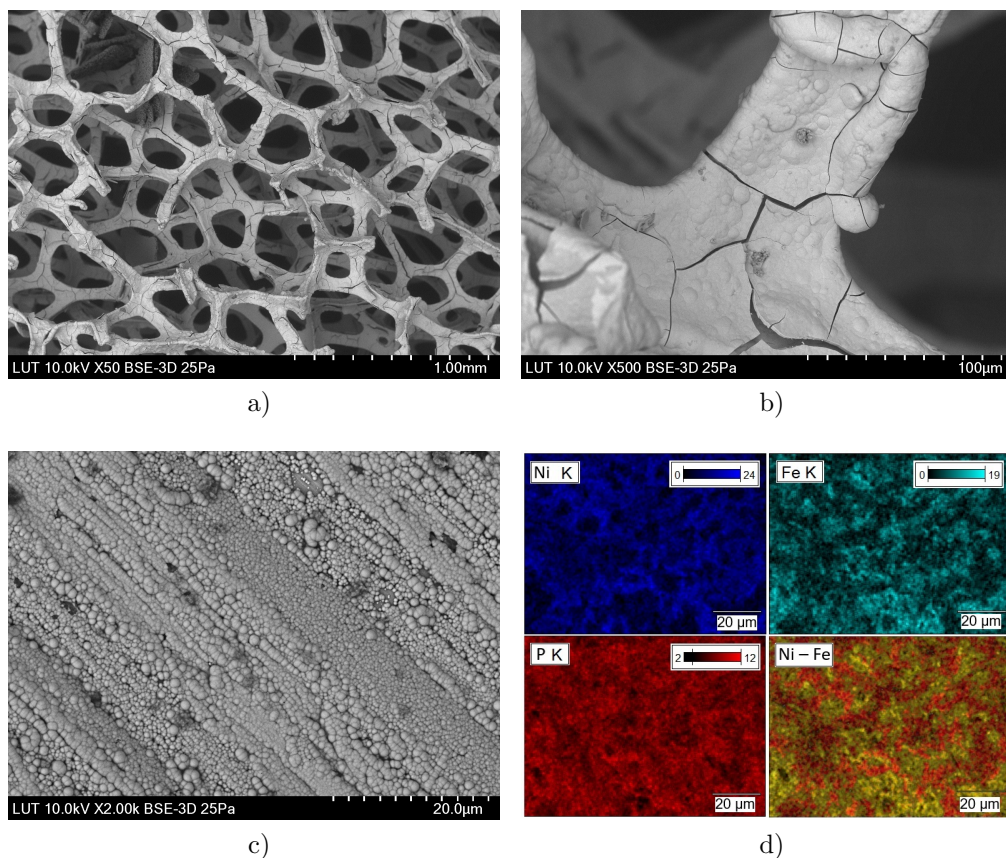


Fig. 6. SEM images of (a–c) NiFe-Pi on NF at different magnifications and (d) the corresponding EDS elemental mappings.

247 The EDXA spectra of bare NF and TM-Pi catalysts are depicted in Fig. 7.
 248 The presence of Ni, Fe, Co, and P can be verified from the corresponding
 249 peaks in the EDXA diagrams. The quantitative EDXA analysis showed that
 250 for Co-Pi catalyst a Co:P ratio varied from 2.5-3:1, while for Fe-Pi a typical
 251 Fe:P indicated a ratio of 1:2.2-2.5. The NiFe-Pi catalysts were prepared
 252 with different Ni/Fe wt.% compositions. The EDXA showed that the weight
 253 percent of Ni and Fe in the NiFe-Pi catalyst deposited onto graphite was
 254 close to the theoretical composition in all cases. For instance, for NiFe-Pi
 255 catalyst a Ni:Fe:P ratio was approximately 8:1:1.6. The amount of Ni was
 256 much higher when using the NF substrate, which can be explained by the
 257 EDXA spectrum probe, which reaches the NF substrate due to penetration

258 through the thin catalyst layer.

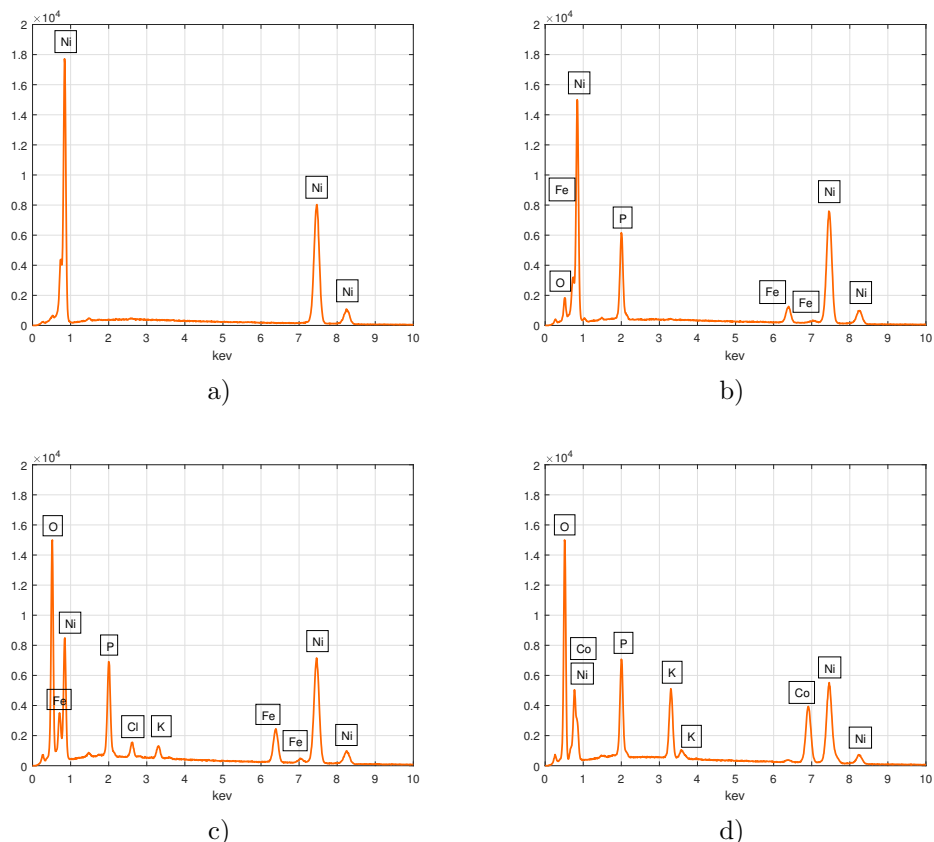


Fig. 7. EDXA spectra for (a) bare nickel foam (NF), (b) NiFe-Pi on NF, (c) Fe-Pi on NF and (d) Co-Pi on NF.

259 3.3. Catalytic activity

260 The electrochemical activity of all TM-Pi catalysts was evaluated using
261 a three-electrode electrolyzer prototype in a neutral PBS electrolyte. In
262 order to find the optimal Ni/Fe composition, NiFe-Pi catalyst deposited onto
263 the graphite substrate with different compositional ratios of Ni and Fe were
264 compared with each other. It was found that electrodes with 25 wt.% of
265 Ni and 75 wt.% of Fe showed the worst electrochemical activity and the
266 highest overpotential. Electrodes with a 50/50 wt.% composition of Ni/Fe

267 had a slightly better OER performance than electrodes with 75/25 wt.%,
268 and thus, the optimal 50/50 wt.% composition was used in subsequent tests.

269 One objective of the study was to investigate the role of the substrate ma-
270 terial in the performance of TM-Pi catalysts. In the present study, graphite
271 was selected as a flat, cheap, and commonly used electrode material, while NF
272 was selected as a substrate with a high specific surface area and good OER
273 activity. Figs. 8a and b show the polarization curves of all TM-Pi catalysts
274 deposited onto graphite and NF at a scan rate of 5 mV s^{-1} in a neutral PBS.
275 The performance of the TM-Pi electrodes was compared with commercial
276 noble materials, represented by Pt and IrO_2 . It can be clearly seen that even
277 though the composition of the catalyst layer is the most important param-
278 eter determining the OER activity, the nature of the substrate onto which
279 the catalyst is deposited also plays a significant role. When studying TM-Pi
280 catalysts deposited onto graphite, the LSV curves revealed that there was no
281 current flow until the potential reached 1.5 V for all the studied electrodes.
282 After 1.5 V, the current at noble Pt and IrO_2 started to rise steeply reach-
283 ing a current density of 1 mA cm^{-2} at overpotentials of 450 mV and 430 mV,
284 respectively. In the case of Co-Pi deposited onto graphite, after 1.5 V, the
285 current increase was less steep, reaching 1 mA cm^{-2} at a significantly higher
286 overpotential of 510 mV. The Fe-Pi and NiFe-Pi catalysts deposited onto
287 graphite were approximately similar in performance, having the onset poten-
288 tial at 1.6 V and reaching current densities of 0.6 mA cm^{-2} and 0.88 mA cm^{-2}
289 at 1.8 V, respectively.

290 The electrocatalytic OER activity of the series of TM-Pi was consider-
291 ably enhanced when deposited onto the NF substrate. As shown in Fig. 8b,
292 all the TM-Pi catalysts exhibited an onset overpotential comparable with
293 the commercial Pt and IrO_2 electrodes. The reference current density of
294 1 mA cm^{-2} was reached at overpotentials of 420 mV, 495 mV, and 534 mV
295 for Co-Pi, NiFe-Pi, and Fe-Pi, respectively. Notably, the OER performance
296 of the Co-Pi catalyst deposited onto the NF surpassed that of the noble elec-
297 trodes during the whole water splitting process. The superior OER activity
298 of Co-Pi is due to both the high active surface area and the high intrinsic
299 catalytic effect. NiFe-Pi and Fe-Pi were once again similar in performance;
300 however, when the potential reached 1.7 V, the activity of NiFe-Pi increased
301 significantly, and the current density reached 4.45 mA cm^{-2} at 1.8 V, exceed-
302 ing that of the Pt and IrO_2 electrodes. The current density of Fe-Pi/NF was
303 about 2.8 times as low as that of NiFe-Pi/NF and almost 3.4 times as low as
304 that of Co-Pi/NF at the end of the electrolysis process.

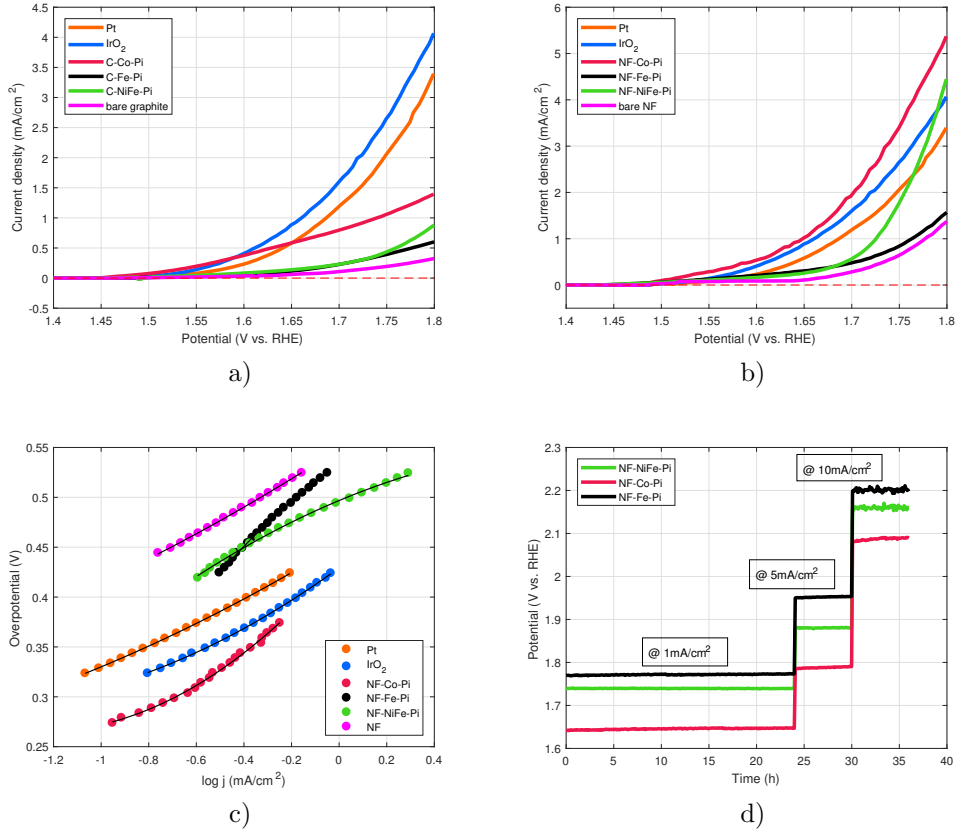


Fig. 8. OER catalytic properties of different catalysts in a PBS with pH=7: (a) LSV curves of TM-Pi deposited onto graphite vs. noble catalysts, (b) LSV curves of TM-Pi deposited onto NF vs. noble catalysts, (c) corresponding Tafel plots and (d) long-term stability tests of TM-Pi.

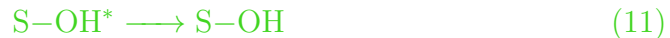
305 In general, the performance of the TM-Pi-containing electrodes is far bet-
 306 ter than that of the bare substrates, especially for the NF series. Although
 307 the coating on all the studied electrodes has the same geometric surface area
 308 of 2.6 cm², the real active area of the catalyst layer is much higher on the NF
 309 substrate because of its unique morphology (Figs. 5a–c). The greater active
 310 area explains the high electrocatalytic activity once the TM-Pi electrocata-
 311 lysts are loaded on the NF. In light of the higher electrocatalytic activity, the
 312 TM-Pi catalysts prepared on NF were selected for further characterization
 313 and evaluation of the OER performance in the present study.

The Tafel slope is an essential representative parameter for evaluation of OER kinetics. Tafel slopes of TM-Pi/NF were determined from the data shown in Fig. 8c. Analysis revealed that IrO₂ and Co-Pi exhibited almost similar Tafel slopes of approximately 60 mV/dec. This value is representative for a chemical rate-determining step in which a fast OER occurs with the OH surface species rearrangement through a surface reaction [46]. Other tested catalysts showed exceptionally high Tafel slopes close to or higher than 120 mV/dec, which is consistent with the comparatively slow OER kinetics characterized by adsorption and discharge of OH⁻ ions as a chemical rate-determining step. Detailed description of kinetic equations under different reaction conditions can be found in [47] while the mechanistic schemes are well-presented in [12]. In general, the mechanism for OER in neutral and alkaline conditions can be described by the following equations [16]:



314 where S represents active sites on the catalyst while S-OH and S-O stand
315 for adsorbed intermediates.

316 When the Tafel slope is close to 120 mV/dec, reaction described by equa-
317 tion 7 represents a rate-determining step. However, the rate-determining
318 step for Tafel slope of 60 mV/dec is different. It is considered that reaction
319 described by equation 7 is divided into the following steps and step described
320 by equation 11 is the rate-determining step for 60 mV/dec:



321 where S-OH* and S-OH stand for adsorbed intermediates with the same
322 chemical structure, but different energy states.
323

324 Stability is an important parameter in evaluation of catalyst performance
325 in practical applications. In this study, stability measurements for a series of

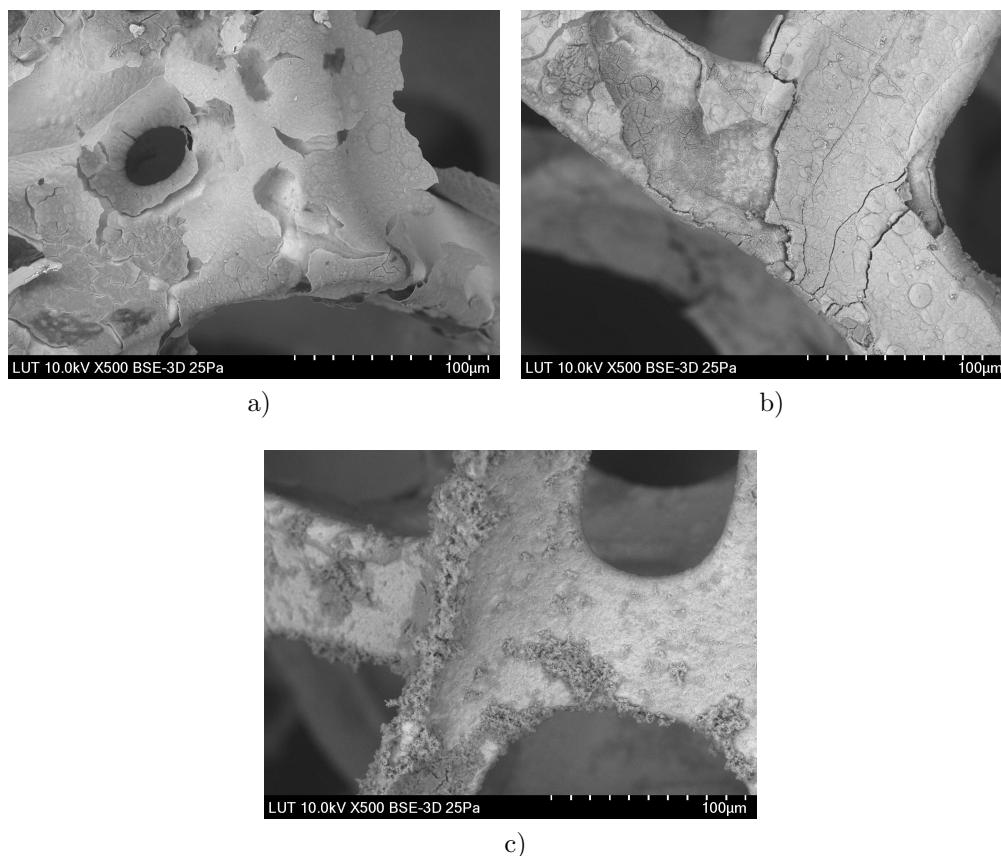


Fig. 9. SEM images recovered after the long-term durability tests for (a) Co-Pi/NF, (b) NiFe-Pi/NF and (c) Fe-Pi/NF.

326 TM-Pi/NF were performed successively at a constant current density of 1,
 327 5, and 10 mA cm⁻² for 36 h in the PBS solution (Fig. 8d). Fig. 9 shows SEM
 328 images of TM-Pi/NF after the 36 h durability test. It can be seen that the
 329 catalyst layer is partly detached from the substrate surface. Nevertheless,
 330 the catalysts continued to maintain a stable potential during the operation
 331 of water splitting. The stable operation indicates that an essential part of
 332 the TM-Pi catalysts remains on the surface, and its amount is sufficient to
 333 maintain the same level of activity. The formation of bubbles on the electrode
 334 surfaces under vigorous oxygen evolution and **partial loss of effective catalyst**
 335 may explain the minor increase in the overpotential over time for all the
 336 studied catalyst materials.

337 *3.4. Overvoltage analysis*

338 In the present study, a simplified cell model was used to describe the
 339 cell voltage components as a function of current density for various TM-Pi
 340 deposited onto NF in a pH-neutral PBS. To apply a developed model, the
 341 voltage and current were measured with the same electrolyzer prototype but
 342 using a two-electrode configuration to measure the whole cell. Cell voltage as
 343 a function of current density is presented in Fig. 10a. Again, NF coated with
 344 an appropriate TM-Pi catalyst was used as a working electrode, the Pt plate
 345 acting as a counter and reference electrode simultaneously. The parameters
 346 U_{rev} , σ_{m} , α , and i_0 in Eq. (6) were determined using experimental voltage
 347 and current data and the method of nonlinear least square regression, and
 348 presented in Table 1. Further, the reversible voltage, the ohmic voltage, and
 349 the activation voltage terms are presented separately in Figs. 10b–d.

Table 1. Experimentally fitted parameters of the simplified cell model with various TM-Pi catalysts deposited onto NF as anodes.

Anode material	U_{rev} (V)	σ_{m} (S cm ⁻¹)	α (-)	i_0 (A cm ⁻²)
NF-NiFe-Pi	1.5799	0.0220	0.1801	0.0001
NF-Co-Pi	1.5166	0.0223	0.1708	0.0001
NF-Fe-Pi	1.5627	0.0222	0.2180	0.0001

Table 2. Reversible voltage, ohmic overvoltage, and activation overvoltage at reference current density of 10 mA m⁻² for various TM-Pi catalysts deposited onto NF.

Anode material	U_{rev} (V)	U_{ohm} (V)	U_{act} (V)	U_{cell} (V)
NF-NiFe-Pi	1.5799	0.2500	0.8294	2.6593
NF-Co-Pi	1.5166	0.2466	0.7866	2.5498
NF-Fe-Pi	1.5627	0.2477	1.0039	2.8143

350 Fig. 10 shows clearly that the measured data are in good agreement with
 351 the experiments with the three-electrode setup. The obtained Co-Pi/NF
 352 exhibited the lowest U_{rev} (Table 1) and outperformed NiFe-Pi/NF and Fe-
 353 Pi/NF throughout the water splitting process. The ohmic overpotential (σ_{m}),
 354 which is mainly caused by the voltage drop across the PBS electrolyte, was

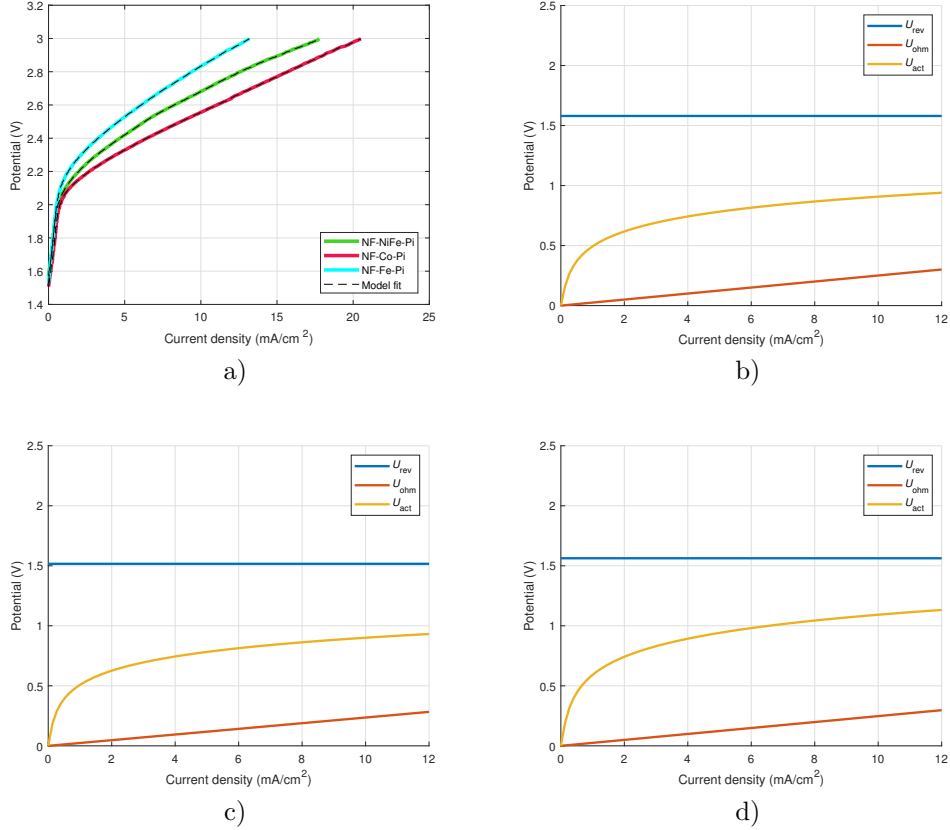


Fig. 10. Overvoltage analysis of the TM-Pi catalysts: (a) cell voltage as a function of current density and the fitted simplified mathematical model; reversible voltage, ohmic overvoltage, and activation overvoltage as a function of current density for (b) NiFe-Pi onto NF, (c) Co-Pi onto NF and (d) Fe-Pi onto NF.

355 approximately similar for all the tested TM-Pi catalysts. The reversible
 356 voltage, ohmic overvoltage, and activation overvoltage at reference current
 357 density of 10 mA cm^{-2} for various TM-Pi catalysts deposited onto NF are
 358 summarized in Table 2. The obtained values match well with the experimen-
 359 tally measured conductivity of the PBS of 22.5 mS cm^{-1} which supports the
 360 use of the simplified cell model.

361 4. Conclusions

362 A comparative study of the OER activity, stability and overvoltage com-
363 ponents of amorphous TM-Pi catalysts deposited by different strategies onto
364 different substrates was presented in this paper. The experimentally ob-
365 served intrinsic OER activities for the studied catalysts in a neutral PBS
366 decreased in the order of Co-Pi > NiFe-Pi > Fe-Pi. The catalysts exhibited
367 activity comparable with or exceeding that of commercial precious Pt and
368 IrO₂ when deposited onto a substrate with a high specific surface area such
369 as NF. The reference current density of 1 mA cm⁻² was reached at relatively
370 small overpotentials of 420 mV, 495 mV, and 534 mV for Co-Pi, NiFe-Pi, and
371 Fe-Pi, respectively. A simplified cell model was used to evaluate the current-
372 voltage characteristics of the studied TM-Pi catalysts and to separate the
373 reversible voltage, the ohmic voltage, and the activation voltage terms. All
374 the studied TM-Pi catalysts exhibited sufficient stability for the OER during
375 the 36 h durability tests in the electrolyzer prototype. The collected data
376 and the proposed simplified cell model are valuable for further studies and
377 evaluation of TM-Pi electrocatalyst performance.

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